



# I: Electronic structure calculations of ordered cubic-based Mg-Li alloys

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#### II: Microstructural evolution of $\alpha+\beta$

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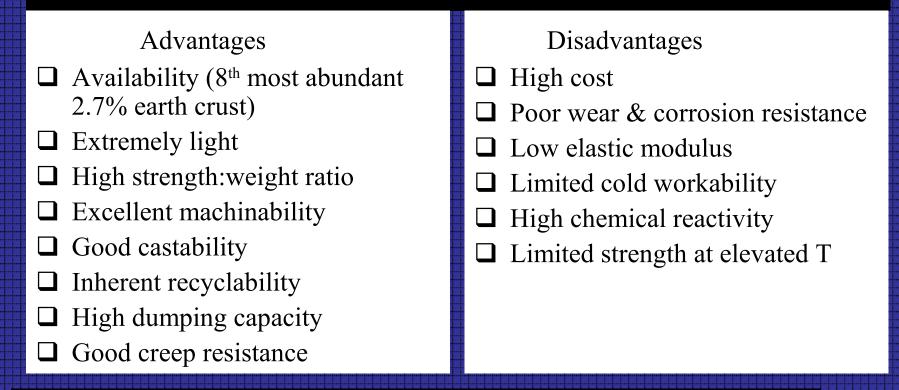
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# Background

Mg alloys have, in the past decade, received revolutionary attention which emanated from the need of lightweight materials in transportation and allied industries, wherein intrinsic strength to weight ratio is of paramount importance.		
Al, Li, Zn, Mn, etc., identified as suitable main alloying elements candidates.		
Due to atomic size, mobility, solute solubility.		
Due to their properties, with lightness and recycleability in the forefront, they are potential candidates to replace steel and aluminium alloys in many structural and mechanical applications.		
Lithium at a density of 0.53 g/cm³ not only enjoys superiority lightness, but also a useful phase change to bcc when alloyed at about 11wt% concentration and increase in ductility.		
High amount of experiment work been carried out on Mg-Li alloys, few investigations has been done through theoretical <i>ab initio</i> techniques.		

#### Mg & Mg alloys: Properties

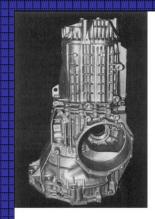


Need: Weight reduction in transport without sacrificing structural strength

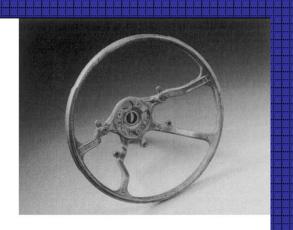
Applications: automotive & aerospace components, computer parts, mobile phones, sporting goods, handheld tools, household equipments

Other possible Appl.: implants due to low weight & inherent biocompatibility

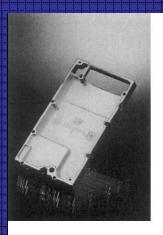
#### Applications



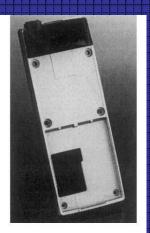




Steering Wheel Frame



Screen Housing



Units For Mobile Handset

#### Mordike et al Fuel

Fuel variable compressionn variable valve control throttle choke losses

< 10%

Fuel qualitiy 1-3%

Diesel direct injection turbo- engine < 15%

Motor/Gear friction

< 1%

Motor/Gear management

< 10%

Estimated potential
Measures are not
additive

Weight light materials < 30%

c<sub>w</sub>-value < 1%

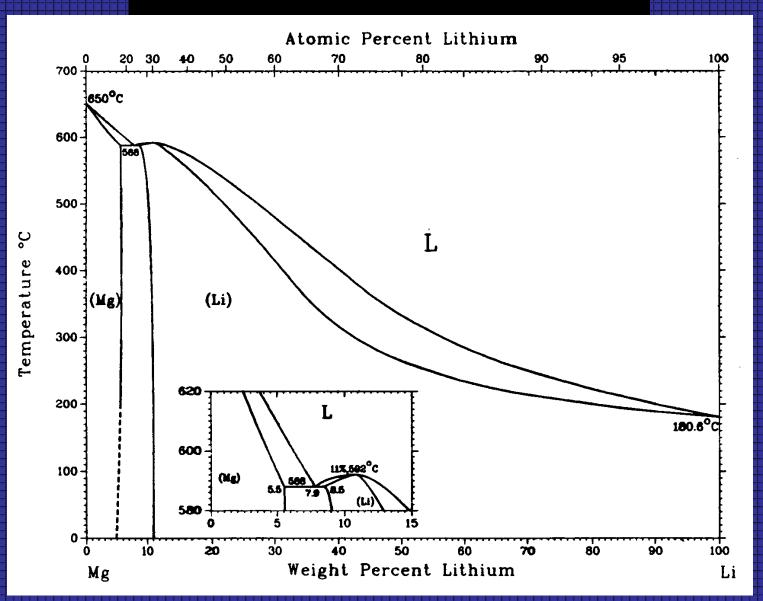
Motorpreheating 2-8%

Efficiency and performance of ancillary aggregates

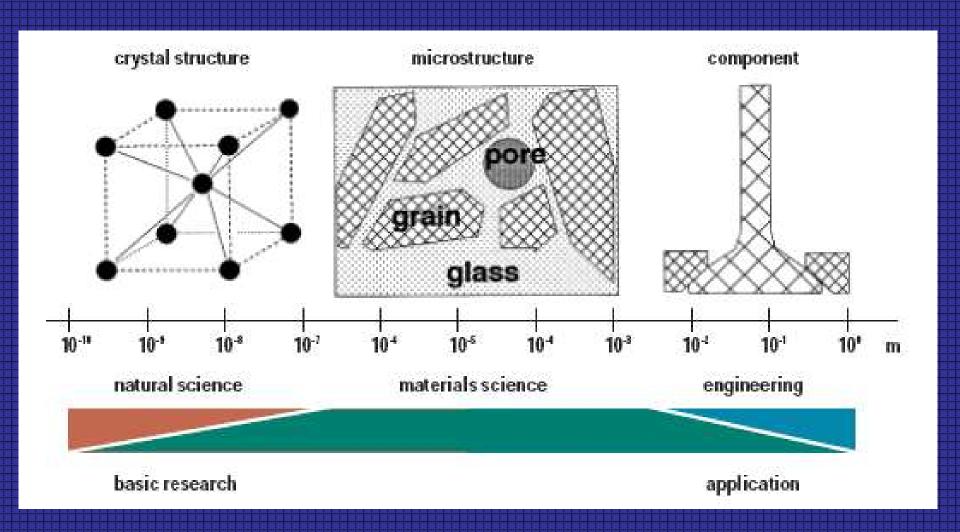
Tyre/road

< 1%

#### Mg-Li Phase diagram



## Hierarchies of length scales



#### I:Methodology

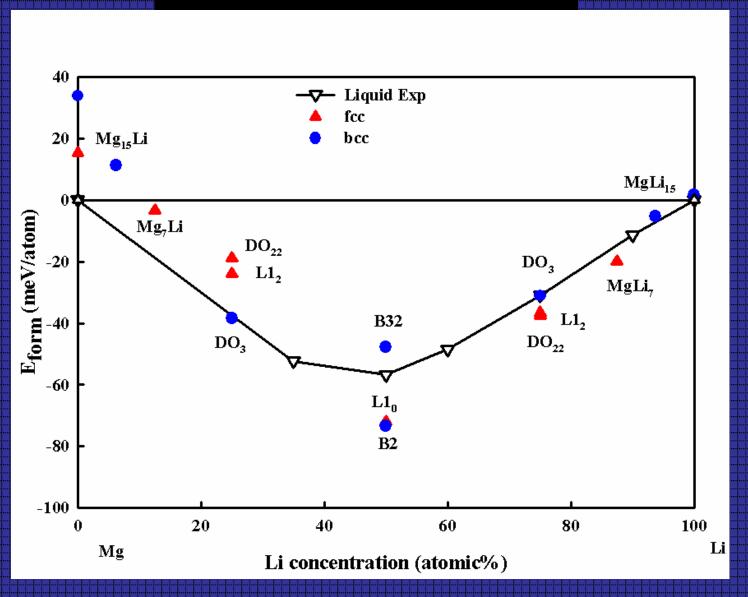
☐ Total energy code, CASTEP, based on DFT within GGA-PBE.

□ Vanderbilt ultrasoft pseudopotentials -using plane wave

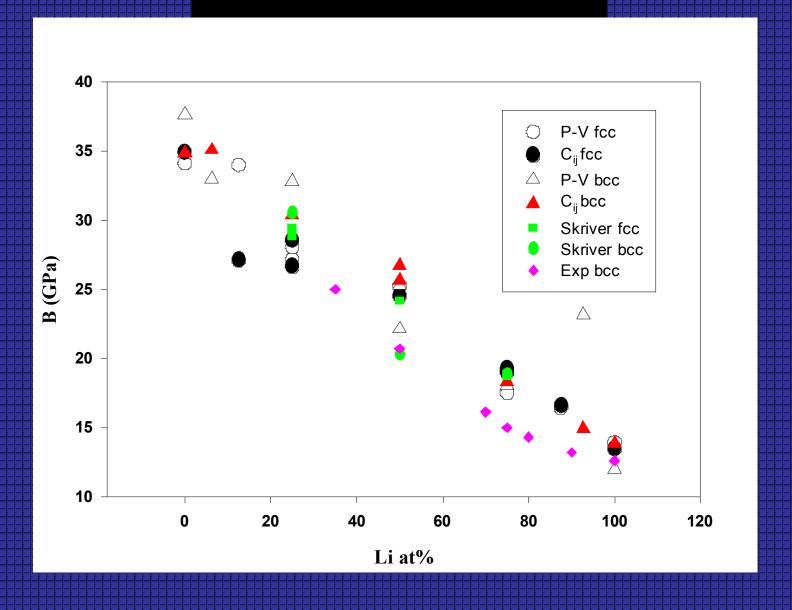
■Monkhorst-Pack scheme

☐ Energy cut-off & k-points were converged

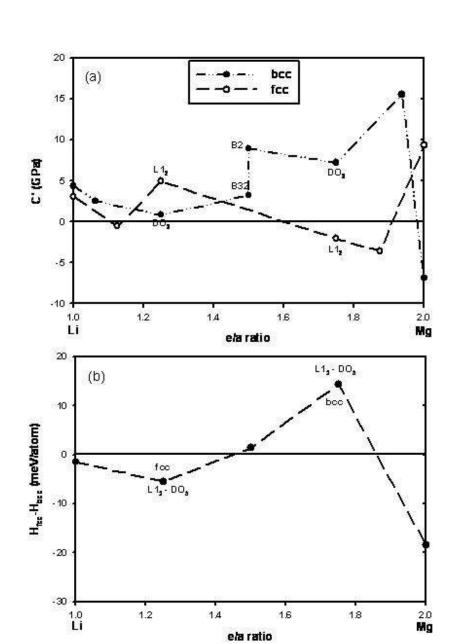
# Results Heats of formation

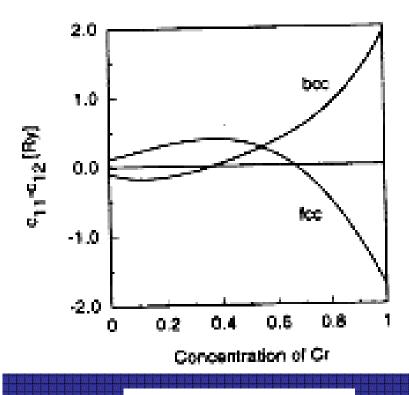


#### Bulk Modulus



#### Shear, Energy correlation





Craievich et al 1996

#### Conclusion

- ☐ Bulk moduli decrease monotonically with increase in Li concentration.
- Predicted heats of formation for all the different ground state superstructures result in a representative stability profile, which shows that the DO<sub>3</sub>, B2 and DO<sub>22</sub> structures are the most stable amongst various phases having Mg<sub>3</sub>Li, MgLi and MgLi<sub>3</sub> compositions, respectively. The similar stability is also being investigated by the use of density of states (DOS).
- ☐ Stability profile hcp-bcc-fcc-hcp is predicted
- $\square$  Elasticity studies predicts stability of DO<sub>3</sub> structure in the  $\alpha + \beta$  region
- $\square$  Correlation C' vs e/a &  $\triangle E$  vs e/a, region where bcc is very stable compared to fcc, the shear modulus is positive for bcc but negative for fcc (i.e. the fcc lattice is the mechanically unstable) and vice versa.

# II: Microstructural evolution of α+β region of MgLi alloys

- □ From ab initio predictions, exp. were conducted as follows
  - Preparations of Mg-Li samples
  - Heat treatments
  - Microstructural phase observations
  - Mechanical tests
  - Structure evolution (OOFEM)

### Sample Preparation Conditions

Arc Melting furnace

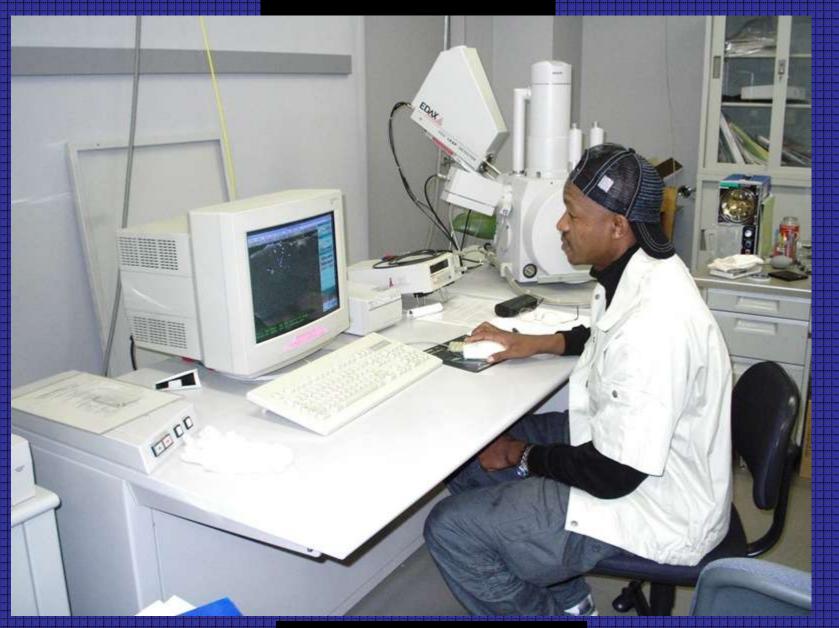
Argon - protecting atmosphere



# Composition of samples

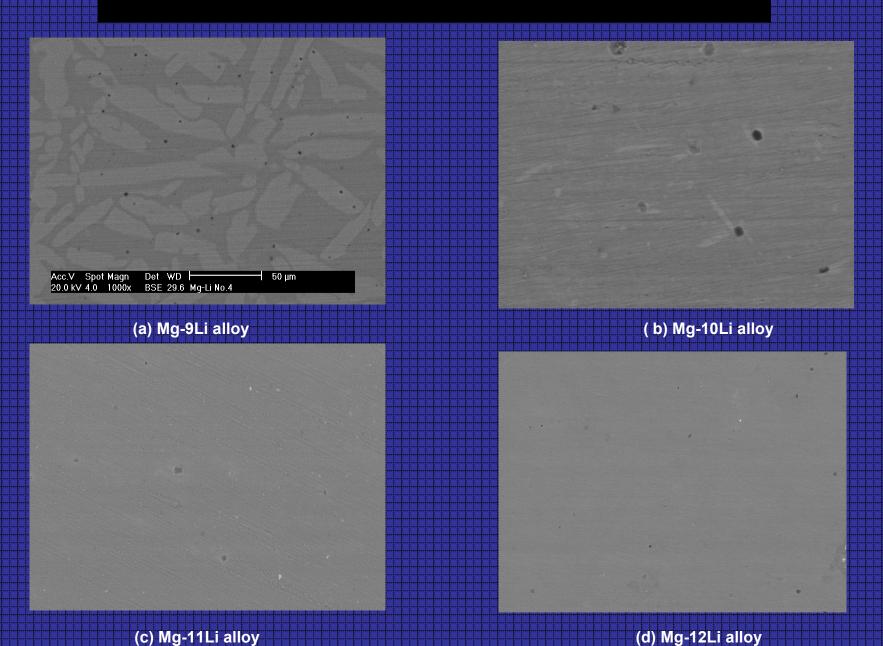
Sample	Li content	Relative ρ
	(wt.%)	g/cm³
Mg-9Li	8.69	1.44
Mg-10Li	10.03	1.40
Mg-11Li	10.99	1.37
Mg-12Li	12.01	1.35

#### SEM



PHILIPS XL 30, Au coater

## SEM Observations:RT



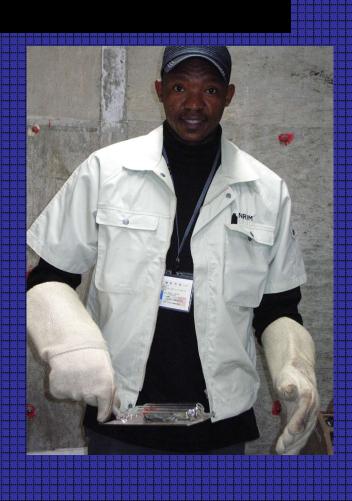
# Microstructure Analysis

Sample	Phases present	Observation
Mg-9Li	α+β (hcp+bcc)	Evenly distributed dual phase alloy
Mg-10Li	α+β (hcp+bcc)	Dominant β phase with α precipitates
Mg-11Li	β (bcc)	Single phase with slight oxidation
Mg-12Li	β (bcc)	Single phase

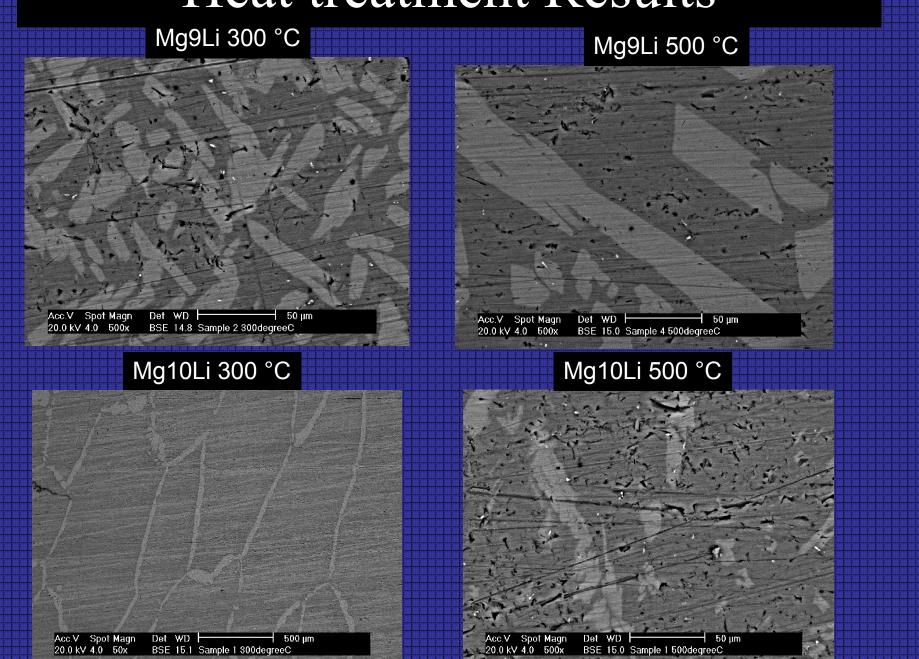
#### Heat treatments



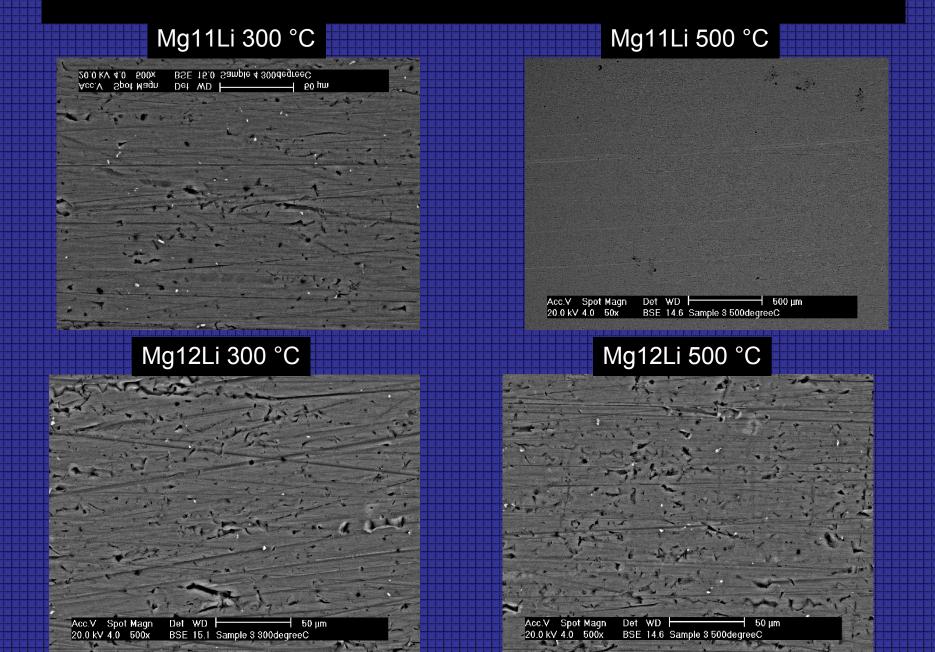




#### Heat treatment Results



#### Heat treatment Results



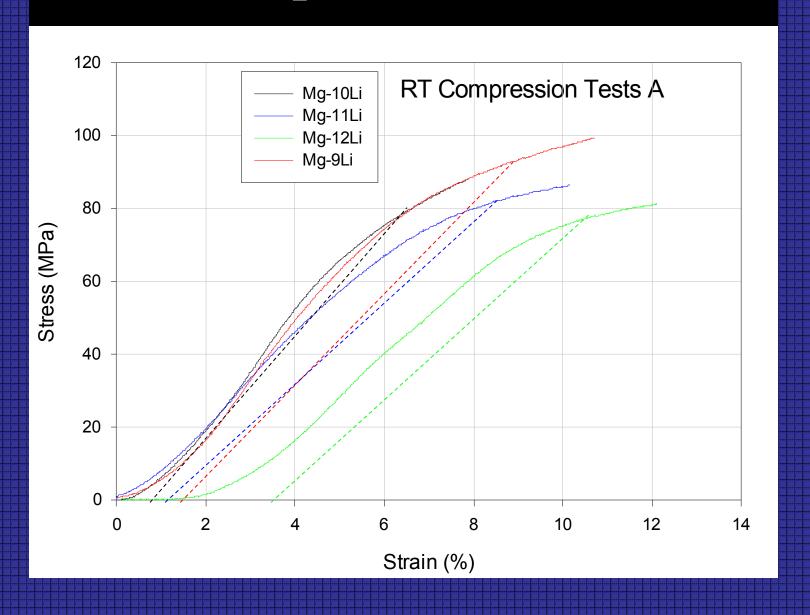
## Compression Tests





Instron machine – cylindrical specimens d=3.87 mm l=6.80 mm Compression/deformation speed = 0.001 mm/s

### Compression Tests



#### III: Future work - Bridging length-scales

#### **Multiscale Materials Design**

Ab initio Calculations (CASTEP, VASP)

- $\Box$  Binaries Mg-X (X = Al, Zn, Si,)
- ☐ Ternaries MgLi-X (X = Al, Zn, Cd, Ti, Ca, Pb, Mn)

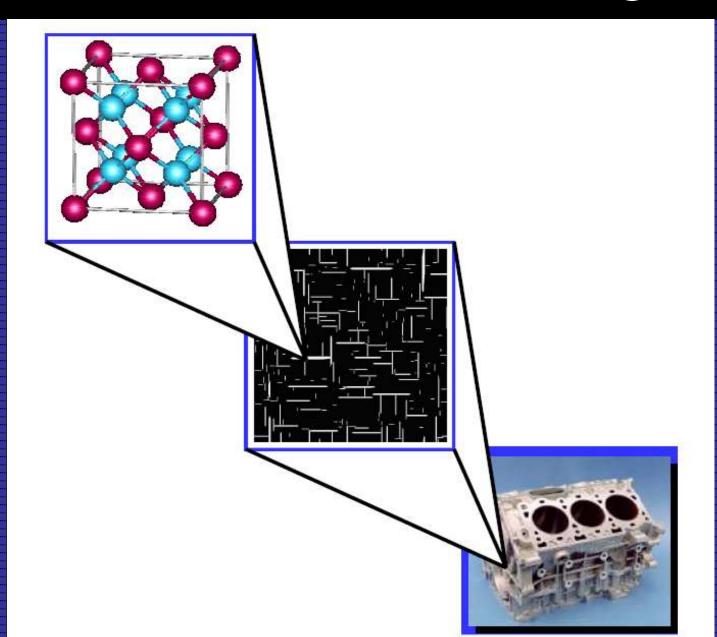
MgZn-X, MgAl-X (X = Li, Zn, Cd, Ti, Mn, RE)

☐ Solid phase transformation Phase diagram predictions Thermodynamic Calculations (CALPHAD, Thermo-Calc)

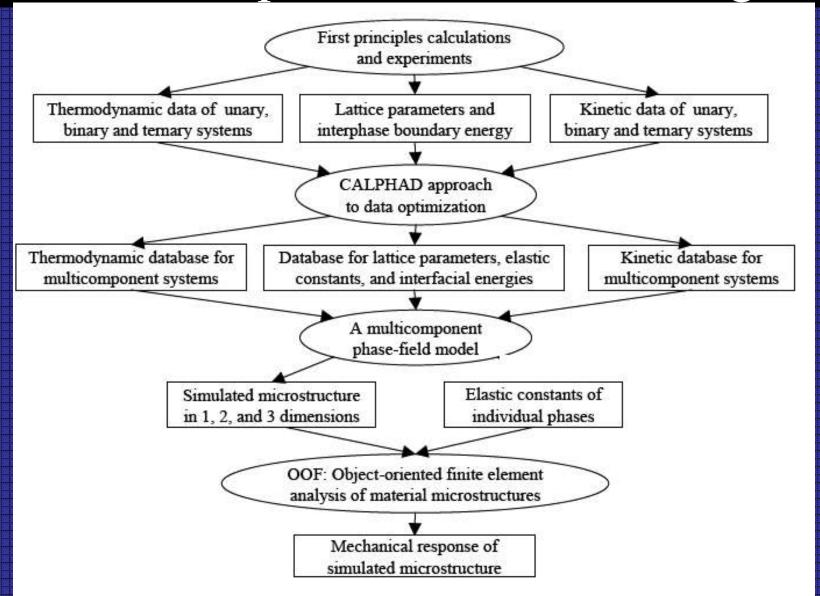
**Phase-field Model (Microstructure)** 

**Microstructure Evolution (OOF)** 

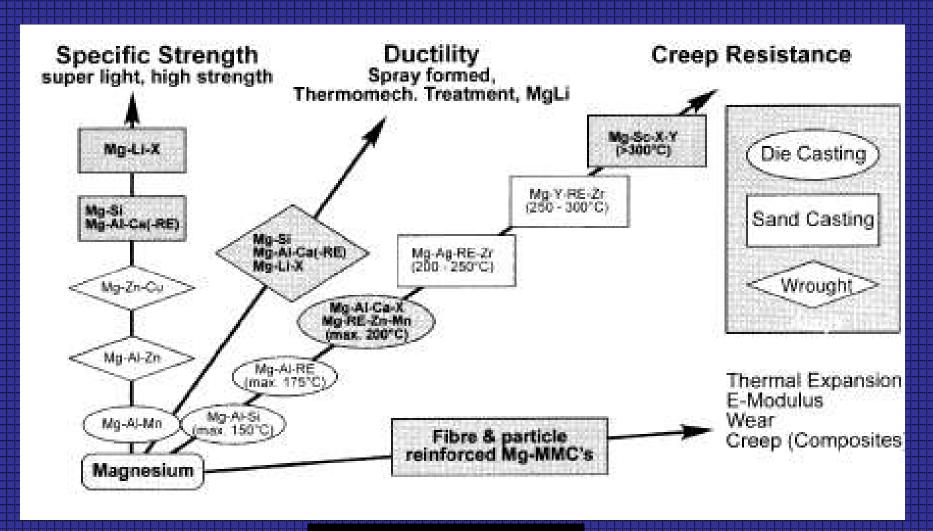
# Multiscale Material Design



# Integrated set of Comp tools for Multicomponent Materials Design



#### Directions of Mg alloy development



Mordike et al - Germany

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