

SnO₃/(OLC) anode for lithium-ion battery applications

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INTRODUCTION

SnO₂ emerged as a promising anode material due to its low cost, safety, high theoretical capacity (782 mA h g⁻¹). However, it experiences large volume changes (~250% or larger) during charge and discharge cycles that cause rapid capacity fade and poor cyclability, which limits its practical use as an anode material. To solve this challenge, one option is utilising effective nanostructures such as nanorods and nanowires. one-dimensional SnO₂ nanorods (NRs) benefit from their favourable geometry that enables them to have direct channels for efficient electron transport. Unfortunately, the direct exposure of SnO₂ nanorods to the electrolyte still cause capacity fading due to the rapturing and unstable nature of the SEI layer, causing reduced coulombic efficiency, high ionic resistance and low electronic conductivity.

In this work, the electrochemical performance of SnO₂ nanorods anchored on onion like carbon (OLC) structure (OLC-derived from the nano-diamonds with diameter of 5-10 nm) were synthesised via microwave irradiation technique. The effect of SnO₂-OLC composites before and after annealed was studied and proved that our SnO₂ integrated OLC exhibits an excellent stable capacity cycling behaviour compared to various other carbon composites reported to date.

Aim

- To utilize nanostructured approach to enhance lithium kinetics.
- To use carbon composite to stablize SEI layer and columbic efficiency.

Materials & Methods

SnO₂ nanorod and SnO2/OLC nanocomposite synthesis

Tin (II) chloride dihydrate (5.65 gm, SnCl₂·2H₂O, Sigma Aldrich, purity 99%) and 0.99 gm of NaOH (Sigma Aldrich, purity 99%) were dissolved in 25 ml of deionized water. The solution subjected to a microwave reaction using Anton Paar Multiwave 3000. The microwave set at constant temperature of 200 °C with 400 W for 20 minutes while stirring.

For SnO₂-OLC, the appropriate amounts of OLC (0.1 g) was dispersed in deionized water by using sonication and add the above raw materials into the solution and the rest of the process is same as above. The dried samples were then annealed at 350 °C at the heating rate of 29min, for 3 h under Argon. The as prepared samples were named herein as SnO₂, SnO₂-OLC and the annealed samples are SnO₂-350°C, SnO₂-OLC 350°C, respectively.



Structure, Morphology and electrochemical analysis

The structure investigated by X-ray diffraction, and lattice parameters derived from the TOPAS-3 software. The SEM images obtained using FE-SEM, JEOL-7600F, and TEM and HR-TEM images obtained from a JEOL-Jem 2100 microscope.

1 M LiPF₆, EC: DMC (1:1) used as electrolyte and Li foil as counter/reference electrode. The electrodes prepared using mixture of an active material, Super P carbon and PVDF binder in the weight ratio of 70:15:15. The electrodes were assembled using a CR2032 coin cell and finally crimped to get a two electrode cell. The assembled coin cells were galvanostatically cycled at 25°C between 0.005 and 1.5 V in a Maccor 4000 battery tester.

Cyclic voltammetry and electrochemical impedance spectroscopy (EIS) were carried out using a Bio-logic VMP3 potentiostat/galvanostat controlled by EC-lab v10.40 software. EIS measurements were obtained with the Autolab Frequency Response Analyser (FRA) software at a frequency range between 100 kHz and 10 mHz with a perturbation amplitude (rms value) of the ac signal of 10 mV. The impedance data were analyzed using Z-view software (version 2.2, Scribner Assoc., Inc., USA).

RESULTS AND DISCUSSIONS

□ Diffraction peaks of all the samples were matched to SnO₂ rutile structure with respect to the JCPDS pattern (No. 41-1445 and a = 4.738 Å and c = 3.187 Å).

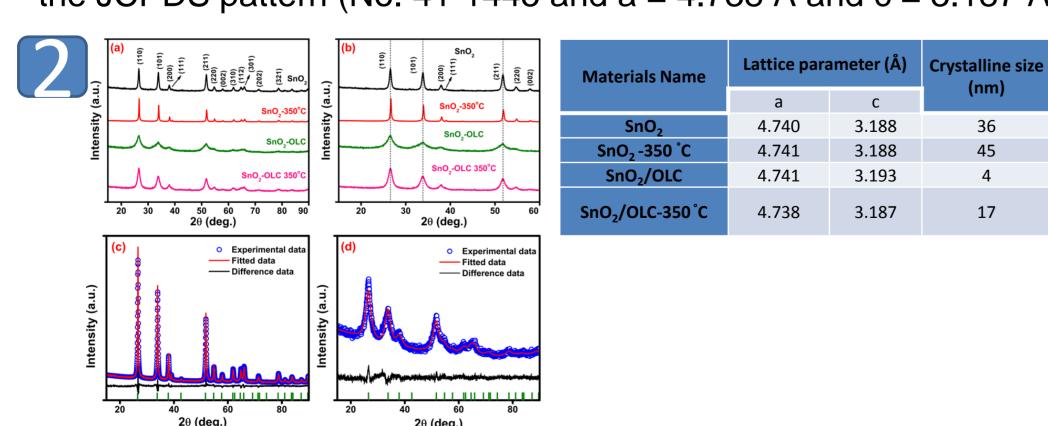
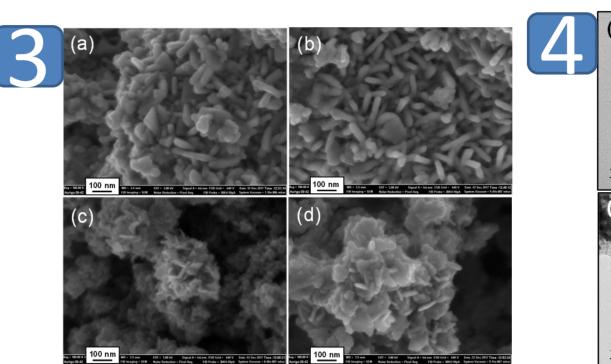


Fig. 2 (a) XRD pattern for all the samples, (b) Enlarged XRD pattern of all the samples between 20° to 40° and Rietveld analysis of (c) SnO₂ nanorod and (d) SnO₂ nanorod-OLC 350 °C composite.

REFERENCES

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350°C, and inserts SAED patterns.

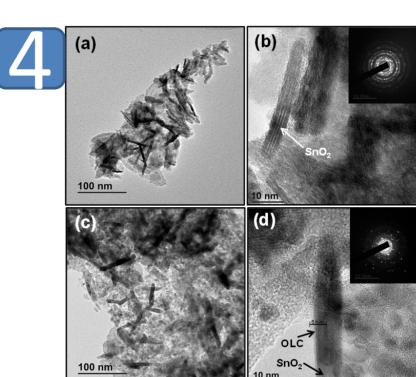


Fig. 3 (a-d) FE-SEM images of the samples (a) SnO₂, (b) SnO₂-350 °C, (c) SnO₂-OLC, (d) SnO₂-OLC 350 °C. Fig. 4 TEM and HR-TEM images of the annealed materials; (a, b) SnO₂-350°C, (c, d) SnO₂-OLC

HR-TEM in Fig 4d shows the presence of OLC ring type structure which successful

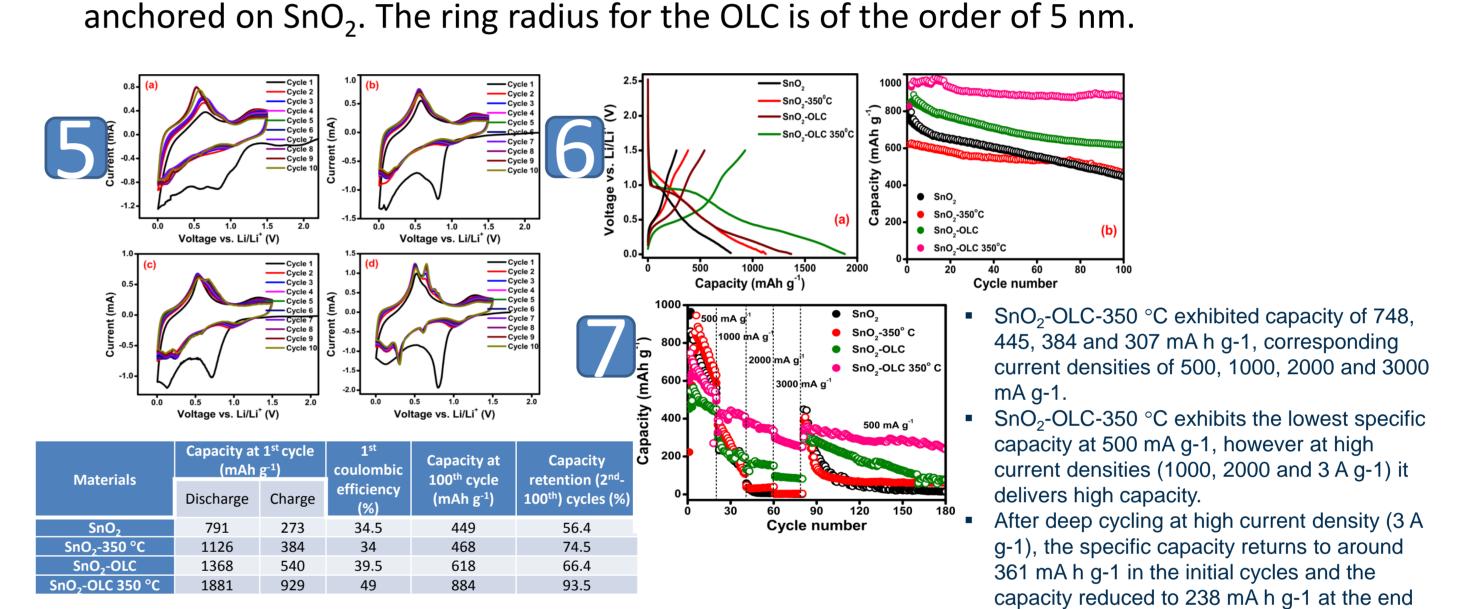


Fig. 5: CV of SnO₂ (a), SnO₂-350 °C (b), SnO₂-OLC (c) and SnO₂-OLC 350 °C (d).; Fig. 6: First cycle of voltage vs. capacity (a) and cycling performance (b) graphs for all samples, cycled at 100 mA g⁻¹.; Fig. 7: Rate performance of SnO₂ and SnO₂-OLC and calcined at 350 °C with the potential window of 0.005-1.5 V.

- $SnO_2 + 4Li^+ + 4e^- \rightarrow 2Li_2O + Sn$
- Sn + xLi⁺ + xe⁻ \leftrightarrow Li_xSn (0 \le x \le 4.4)

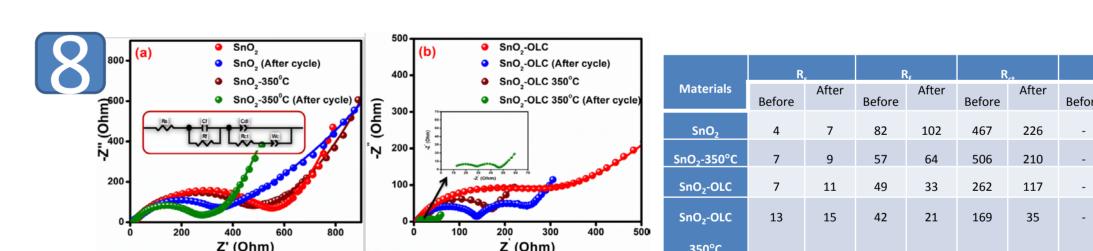


Fig. 8: Nyquist plots SnO₂, SnO₂ 350 °C (a) SnO₂-OLC, SnO₂-OLC 350 °C (b) and Equivalent circuit for all the compounds shown in inset (a) for before and after 100 cycling.

CONCLUSIONS

- Facile and rapid microwave process was implemented to synthesize SnO₂ nanorods anchored on OLC composites as an anode material for LIB applications.
- XRD reveals that, all the composites exhibited tetragonal structure of SnO₂, without any sign of other impurities.
- The OLC coated SnO₂ annealed at 350 °C exhibits highest specific capacity of 884 mA h g⁻¹ at the end of the 100th cycles with the capacity retention of \sim 94%, and improved coulombic efficiency as the OLC serves as barrier layer between the SnO₂ structure and electrolyte to stabilize the SEI.
- Also it exhibits excellent specific capacity at high current rates with good cycling stability, which attributed to the high surface area of OLC and could enhance the electron transportation and high lithium ion diffusion during cycling.
- EIS evidenced that the OLC-SnO₂ composite after cycling showed low resistance and the electron transportation are associated with grain and grain boundary of the materials.